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# $CO_2$ adsorption and activation over $\gamma$ -Al $_2O_3$ -supported transition metal dimers: A density functional study

Yun-xiang Pan a,b, Chang-jun Liu a, Tomasz S. Wiltowski c, Qingfeng Ge b,\*

- <sup>a</sup> Key Laboratory for Green Chemical Technology of Ministry of Education, School of Chemical Engineering, Tianjin University, Tianjin 300072, China
- <sup>b</sup> Department of Chemistry and Biochemistry, Southern Illinois University, 1245 Lincoln Drive, Carbondale, IL 62901, USA
- <sup>c</sup> Department of Mechanical Engineering and Energy Processes and Coal Research Center, Southern Illinois University, Carbondale, IL 62901, USA

# ARTICLE INFO

Article history:
Available online 10 June 2009

Keywords:  $CO_2$  adsorption and activation Transition metal dimers  $\gamma$ - $Al_2O_3$  Density functional theory

# ABSTRACT

Catalytic conversion of  $CO_2$  to liquid fuels has the benefit of reducing  $CO_2$  emission. Adsorption and activation of  $CO_2$  on the catalyst surface are key steps of the conversion. Herein, we used density functional theory (DFT) slab calculations to study  $CO_2$  adsorption and activation over the  $\gamma$ -Al $_2O_3$ -supported 3d transition metal dimers ( $M_2/\gamma$ -Al $_2O_3$ , M=Sc-Cu).  $CO_2$  was found to adsorb on  $M_2/\gamma$ -Al $_2O_3$  negatively charged and in a bent configuration, indicating partial activation of  $CO_2$ . Our results showed that both the metal dimer and the  $\gamma$ -Al $_2O_3$  support contribute to the activation of the adsorbed  $CO_2$ . The presence of a metal dimer enhances the interaction of  $CO_2$  with the substrate. Consequently, the adsorption energy of  $CO_2$  on  $M_2/\gamma$ -Al $_2O_3$  is significantly higher than that on the  $\gamma$ -Al $_2O_3$  surface without the metal dimer. The decreasing binding strength of  $CO_2$  on  $M_2/\gamma$ -Al $_2O_3$  as  $M_2$  changes from  $Sc_2$  to  $Cu_2$  was attributed to decreasing electron-donation by the supported metal dimers. Hydroxylation of the support surface reduces the amount of charge transferred to  $CO_2$  for the same metal dimer and weakens the  $CO_2$  chemisorption bonds. Highly dispersed metal particles maintained at a small size are expected to exhibit good activity toward  $CO_2$  adsorption and activation.

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#### 1. Introduction

Catalytic conversion of carbon dioxide (CO<sub>2</sub>) into liquid fuels has been identified as one of the priority research directions in a recent U.S. Department of Energy Report, "Basic Research Needs: Catalysis for Energy" [1]. Recycling CO2 to make liquid fuels neutralizes CO<sub>2</sub> emissions, thereby alleviating the greenhouse effect caused by CO<sub>2</sub>. However, converting CO<sub>2</sub> to liquid fuel is an energy-intensive process due to the nature of the reactions involved. Carbon dioxide can be used as a reforming reagent for methane to produce synthesis gas  $(CO + H_2)$  [2-11], which can then be converted to liquid fuels. Fe-, Co- and Ni-based catalysts were found to have reasonable activities for CO2 reforming of methane [4–11]. Alternatively, CO<sub>2</sub> can be directly hydrogenated, where the products from direct CO<sub>2</sub> hydrogenation depend on the catalysts used. Over the Ni-based catalysts, methane is one of the main products [12-14]. On the other hand, methanol will be produced predominantly over the Cu-based catalysts [15,16]. In these heterogeneous CO<sub>2</sub> conversion processes, adsorption of CO<sub>2</sub> and activation of the C-O bond are key steps. A detailed characterization of CO<sub>2</sub> interaction with the catalysts at the molecular level will help us to better understand the underlying mechanisms of the reactions and aid in the elucidation of key factors that affect the performance of the catalysts.

There have been many studies on the interaction between CO<sub>2</sub> and transition metal surfaces [17]. On the basis of combined experimental and computational studies, Vesselli et al. [18] suggested that the CO<sub>2</sub> molecule was adsorbed on the Ni (110) surface through the carbon atom in a negatively charged and bent configuration. Wang et al. studied the chemisorption of CO<sub>2</sub> on the Ni (1 1 1), Ni (1 0 0) and Ni (1 1 0) surfaces using DFT methods and reported that the C=O bond of CO<sub>2</sub> was activated due to electron transfer from the Ni surface into the anti-bonding orbital of CO<sub>2</sub> [19]. The C=O bond activation ability of the Ni surfaces was suggested to be in the order of Ni  $(1\ 1\ 0) > Ni (1\ 0\ 0) > Ni (1\ 1\ 1)$ . UV photoelectron spectroscopy, X-ray photoelectron spectroscopy and Auger electron spectroscopy have all been applied to characterize CO<sub>2</sub> adsorption on potassium-covered Fe (110) surface at 85 K [20,21]. A small amount of CO2 was found to dissociate into CO and oxygen whereas the molecularly adsorbed CO<sub>2</sub> was found in a bent configuration.

The mechanisms for  $CO_2$  adsorption and activation over the oxide-supported metal catalysts are expected to be different from that on the pure metal surfaces due to the presence of the oxide support. Aksoylu and Önsan [14] investigated  $CO_2$  adsorption and methanation over oxide-supported nickel catalysts. They found

<sup>\*</sup> Corresponding author. Tel.: +1 618 453 6406; fax: +1 618 453 6408. E-mail address: qge@chem.siu.edu (Q. Ge).

that the oxide support has no activity for methanation but is active for  $CO_2$  adsorption and activation. de Leitenburg et al. [22] studied the interaction between  $CO_2$  and the ceria-supported late transition metals (Rh, Ru, Pt, Pd and Ir). They reported that  $CO_2$  activation on the surface  $Ce^{3+}$  sites led to the formation of  $CO_2$  followed by the oxidation of  $Ce^{3+}$  to  $Ce^{4+}$ . The presence of oxygen vacancies in ceria created additional reduction potential for the reduction of  $CO_2$  to  $CO_2$  and/or surface carbonaceous species. The carbonaceous species were rapidly hydrogenated to  $CH_4$  over the supported metals. The effect of the support on  $CO_2$  adsorption and activation was also reported over the oxide-supported Rh catalysts. On a clean Rh surface the adsorption of  $CO_2$  is weak and non-dissociative, whereas over the oxide-supported Rh catalysts  $CO_2$  dissociates easily [23,24].

Alumina,  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> in particular, has been commonly used as support in heterogeneous catalysis, including CO<sub>2</sub> conversion. However, a detailed characterization of the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>-supported metal catalysts as well as their activity toward CO2 adsorption and activation is not readily available. This can be partly attributed to the structural complexity of  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> [25–28]. The  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> structural models based on the defective spinel model [25,29] and non-spinel model [27] have been proposed. We employed both types of models in our previous studies [30–33]. In the present study, we chose the non-spinel model based on the fact that its (100) surface exposes the penta-Al sites that were observed experimentally [34,35]. Herein, we used density functional theory (DFT) slab calculations to study the interaction of  $CO_2$  with the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>-supported 3d transition metal dimers (Sc<sub>2</sub>, Ti<sub>2</sub>, V<sub>2</sub>, Cr<sub>2</sub>, Mn<sub>2</sub>, Fe<sub>2</sub>, Co<sub>2</sub>, Ni<sub>2</sub> and Cu<sub>2</sub>). The dimers were used since they are the smallest units that allow us to probe both metal-metal and metalsupport interactions. In fact, features of dimer have been directly imaged on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> [36]. We selected the (1 1 0) surface of  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> to model the support for the transition metal dimers. This is based on the fact that the (1 1 0) surface was estimated to account for 83% of the total surface area of  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> and is expected to dominate the exposed surfaces on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> nanocrystallites [27].

This paper is organized as follows. Computational method and models are described in Section 2. Results and discussion are given in Section 3. Finally, the main conclusions are presented in Section 4. For clarity of presentation, the transition metal dimers will be referred to as  $M_2$  with M = Sc-Cu in the following discussions when a distinction of the metal is not necessary.

# 2. Computational methodology and models

The methodology and the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> model are similar to those used in our previous studies [30,33]. All the calculations were performed by using the Vienna ab-initio simulation package (VASP) [37]. The projector augmented wave method was used to describe the interaction between ions and electrons [38,39] and the PAW potentials of the 3d elements have been used our previous calculations [40]. The non-local exchange-correlation energy was evaluated using the Perdew-Burke-Ernzehof functional [41]. A plane wave basis set with a cutoff energy of 400 eV was used. A  $(2 \times 2 \times 1)$  grid was used to generate k-points according to the Monkhorst–Pack method [42]. Test calculations with a k-point grid of  $(4 \times 4 \times 1)$  gave a numerical difference in  $CO_2$  adsorption energy of less than 0.005 eV. The atomic structures were relaxed using the conjugate gradient algorithm scheme as implemented in the VASP code until the forces on unconstrained atoms were <0.03 eV/Å. Bader charge analyses were also performed for all the adsorption systems [43,44].

The unit cell size of the  $\gamma\text{-Al}_2O_3$  (1 1 0) surface and the number of the Al $_2O_3$  layers were kept the same as in our previous study [33]. The  $\gamma\text{-Al}_2O_3$  (1 1 0) surface was modeled by a supercell with a dimension of (8.40 Å  $\times$  8.07 Å  $\times$  19.18 Å). Twelve Al $_2O_3$  molecular

units in the slab were distributed in six layers. The vacuum region separating the slabs along  $[0\ 0\ 1]$  direction was set to  $12\ \text{Å}$ . In all calculations, the bottom two layers were frozen in their bulk positions, whereas the remaining four layers together with  $M_2$ , the adsorbed  $CO_2$  and/or hydroxyls were allowed to relax.

The binding energy of  $M_2$  with the surface,  $\Delta E_{binding}$ , was calculated as follows:

$$\Delta E_{binding} = -(E_{M_2/\gamma - Al_2O_3} - E_{\gamma - Al_2O_3} - E_{M_2})$$

where  $E_{\rm M_2/\gamma-Al_2O_3}$ ,  $E_{\gamma-Al_2O_3}$  and  $E_{\rm M_2}$  are the total energies of  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> slab with M<sub>2</sub>, the bare  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> slab and the free M<sub>2</sub>, respectively. The adsorption energy of CO<sub>2</sub>,  $\Delta E_{ad}$ , was defined as:

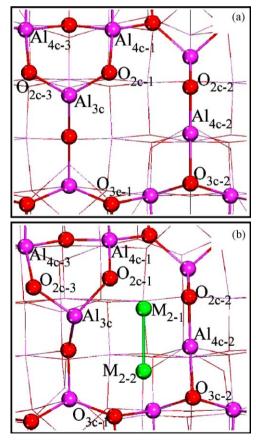
$$\Delta E_{ad} = -(E_{\text{CO}_2-\text{M}_2/\gamma-\text{Al}_2\text{O}_3} - E_{\text{M}_2/\gamma-\text{Al}_2\text{O}_3} - E_{\text{CO}_2})$$

where  $E_{\text{CO}_2-\text{M}_2/\gamma-\text{Al}_2\text{O}_3}$  and  $E_{\text{CO}_2}$  represent the total energies of the slab simulating CO<sub>2</sub> adsorbed on M<sub>2</sub>/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> and the free CO<sub>2</sub> molecule, respectively. The energy of the free CO<sub>2</sub> molecule,  $E_{\text{CO}_2}$ , was computed by placing a single CO<sub>2</sub> molecule in a 10 Å  $\times$  10 Å  $\times$  10 Å cubic box. Consequently, positive values of  $\Delta E_{binding}$  or  $\Delta E_{ad}$  indicate energetically favorable processes according to these definitions of binding energy and adsorption energy.

#### 3. Results and discussion

# 3.1. $M_2$ supported on the dry $\gamma$ - $Al_2O_3$ (1 1 0) surface

The top view of the dry  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (1 1 0) surface was provided in Fig. 1a to label the surface atoms. On the dry surface, 3-fold-coordinated aluminum (Al<sub>3c</sub>), 4-fold-coordinated aluminum (Al<sub>4c</sub>), 2-fold-coordinated oxygen (O<sub>2c</sub>) and 3-fold-coordinated oxygen



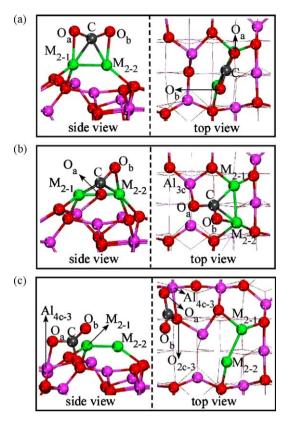
**Fig. 1.** Top view of the dry  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (1 1 0) surface (a) and the schematic structural model of D(M<sub>2</sub>) (b). Color coding: red, O atoms; pink, Al atoms; green, M atoms. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of the article.)

(O<sub>3c</sub>) atoms are exposed, among which Al<sub>3c</sub>, Al<sub>4c</sub> and O<sub>2c</sub> are coordinately unsaturated. When adsorption involves more than one atom of the same type of surface sites, the sites will be distinguished by adding an additional subscript. For example, Sc<sub>1</sub>-O<sub>2c-2</sub> represents the bond formed between the first Sc atom in the Sc dimer and the second  $O_{2c}$  atom at the binding site. Different surface sites were explored for M2 adsorption, and adsorption at these sites resulted in a number of configurations. The most stable adsorption configuration of M<sub>2</sub> among the stable sites was selected as the substrate in the study of CO<sub>2</sub> adsorption and was named as  $D(M_2)$ , with D indicating the dry oxide surface. For example,  $D(Ni_2)$ represents the most stable configuration for Ni<sub>2</sub> supported on the dry surface. A schematic structural model of D(M2) is shown in Fig. 1b. In  $D(M_2)$ , the dimer is located at a site surrounded by two  $O_{2c}$  atoms ( $O_{2c-1}$  and  $O_{2c-2}$ ), two  $O_{3c}$  atoms ( $O_{3c-1}$  and  $O_{3c-2}$ ), an  $Al_{3c}$ atom and an  $Al_{4c}$  ( $Al_{4c-2}$ ) atom. We kept the two metal atoms in dimer form and did not explore the configuration in which metal atoms are separated from each other. The binding energies of M<sub>2</sub> in D(M<sub>2</sub>) as well as the distances between metal and surface atoms are summarized in Table 1.

Among all M<sub>2</sub> from Sc<sub>2</sub> to Cu<sub>2</sub>, Sc<sub>2</sub> exhibits the strongest binding with the dry surface, with a binding energy of 4.65 eV. In D(Sc<sub>2</sub>), the distances between Sc and the surface O atoms,  $Sc_{2-1}-O_{2c-1}$ , 2.00 Å,  $Sc_{2-1}-O_{2c-2}$ , 2.01 Å,  $Sc_{2-2}-O_{3c-1}$ , 2.11 Å and  $Sc_{2-2}-O_{3c-2}$ , 2.06 Å, indicate that bonds were formed between the atom pairs. The binding energy of Ti<sub>2</sub> in D(Ti<sub>2</sub>) is 4.19 eV, which is lower by 0.46 eV than that of  $Sc_2$  in  $D(Sc_2)$ . In  $D(Ti_2)$ , the  $Ti_{2-1}$  atom bonds to the surface  $O_{2c-1}$  and  $O_{2c-2}$  atoms whereas the  $Ti_{2-2}$  atom bonds to the surface O<sub>3c-1</sub> atom only. The bond distances of Ti<sub>2-1</sub>-O<sub>2c-1</sub>, Ti<sub>2-</sub>  $_{1}$ – $O_{2c-2}$  and  $Ti_{2-2}$ – $O_{3c-1}$  are 1.95 Å, 1.96 Å, and 2.20 Å, respectively. As we moved to  $V_2$ , the binding energy decreases significantly, as shown in Table 1. In D(V<sub>2</sub>), V<sub>2</sub> appears to bound with one end of the dimer  $(V_{2-1}-O_{2c-1})$  and  $V_{2-1}-O_{2c-2}$ . Among all  $M_2$ ,  $Cr_2$  shows the weakest binding with the surface, with a binding energy of 2.07 eV, although both ends of Cr<sub>2</sub> interact with the surface, as indicated by the distances:  $Cr_{2-1}-O_{2c-1}$  (1.97 Å),  $Cr_{2-1}-O_{2c-2}$  (1.97 Å),  $Cr_{2-2}-O_{3c-1}$ (2.09 Å) and  $Cr_{2-2}-O_{3c-2}$  (2.07 Å). As shown in Table 1, the binding energy of Mn<sub>2</sub> in D(Mn<sub>2</sub>) is higher by 0.95 eV than that of Cr<sub>2</sub> in  $D(Cr_2)$ . In  $D(Mn_2)$ ,  $Mn_{2-1}-O_{2c-1}$  (1.97 Å),  $Mn_{2-1}-O_{2c-2}$  (1.99 Å) and  $Mn_{2-2}-O_{3c-1}$  (2.04 Å) bonds were formed between  $Mn_2$  and the surface. From Mn<sub>2</sub> to Cu<sub>2</sub>, the binding energy only increases slightly, from 3.02 eV to 3.33 eV. D(Fe<sub>2</sub>), D(Co<sub>2</sub>), D(Ni<sub>2</sub>) and D(Cu<sub>2</sub>) have similar structures in which M2 interacts with the surface through  $M_{2-1}-O_{2c-1}$ ,  $M_{2-1}-O_{2c-2}$  and  $M_{2-2}-O_{3c-1}$  bonds. For example, the three bonds in D(Ni<sub>2</sub>) are Ni<sub>2-1</sub>-O<sub>2c-1</sub> (1.91 Å), Ni<sub>2-</sub>  $_{1}$ -O<sub>2c-2</sub> (2.09 Å) and Ni<sub>2-2</sub>-O<sub>3c-1</sub> (2.05 Å).

# 3.2. $CO_2$ adsorption and activation on $D(M_2)$

Different  $CO_2$  adsorption configurations on  $D(M_2)$  were explored and three adsorption configurations  $\eta_1,\,\eta_2$  and  $\eta_3$  were found to be stable. In this section, the structural properties of these adsorption configurations will be described first. For clarity, the



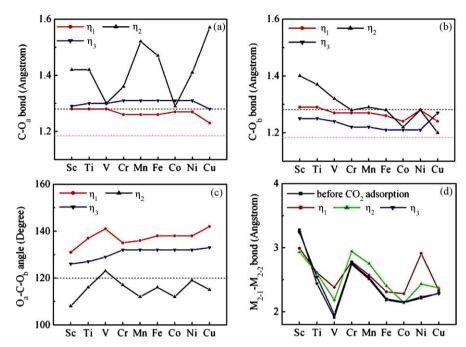
**Fig. 2.** Schematic structural models of adsorption configurations  $\eta_1$  (a),  $\eta_2$  (b) and  $\eta_3$  (c). Color coding, grey, C atoms; others are the same as in Fig. 1.

oxygen atoms of  $\text{CO}_2$  were labeled as  $\text{O}_a$  and  $\text{O}_b$  if the two atoms were located at unequivalent positions.

A schematic structural model for  $\eta_1$  configuration is shown in Fig. 2a. In  $\eta_1$ , CO<sub>2</sub> interacts with D(M<sub>2</sub>) through C-M<sub>2-1</sub>, C-M<sub>2-2</sub>,  $O_a-M_{2-1}$  and  $O_b-M_{2-2}$  bonds. The adsorbed  $CO_2$  is in a geometry that was greatly distorted from the free CO<sub>2</sub> molecule. In Fig. 3, the variations of the C-O<sub>a</sub> and C-O<sub>b</sub> bond distances and the O<sub>a</sub>-C-O<sub>b</sub> angle were plotted against the metals. The structural parameters of the free CO<sub>2</sub> molecule and CO<sub>3</sub><sup>2-</sup> ion were also plotted in the figure for references. Both the C-O<sub>a</sub> and C-O<sub>b</sub> bonds were elongated from the C-O bond in free CO<sub>2</sub> and the  $O_a$ -C- $O_b$  angle was reduced significantly in  $\eta_1$  from 180°. For instance, in the  $\eta_1$  configuration on D(Ni<sub>2</sub>), the C-O<sub>a</sub> and C- $O_b$  bonds are 1.27 Å and 1.28 Å, respectively, and the  $O_a$ -C- $O_b$ angle is 138°. Elongation of the C-O bonds and reduction of the O-C-O angle indicate the partial activation of CO<sub>2</sub> [19]. The variation of the M<sub>2-1</sub>-M<sub>2-2</sub> bond distance before and after CO<sub>2</sub> adsorption is also shown in Fig. 3d. Clearly, the adsorption of  $CO_2$  in  $\eta_1$  configuration on  $D(M_2)$  made the  $M_{2-1}-M_{2-2}$ (M = Ti - Cu) bonds stretch. Only the  $Sc_{2-1} - Sc_{2-2}$  distance was reduced slightly.

**Table 1** Distances (in Å) between metal and surface atoms and binding energies (in eV) of  $M_2$  in  $D(M_2)$  (M = Sc–Cu).

	$M_{2-1}-O_{2c-1}$	$M_{2-1}-O_{2c-2}$	$M_{2-1}$ - $Al_{4c-2}$	$M_{2-2}$ - $O_{3c-1}$	$M_{2-2}$ - $O_{3c-2}$	$M_{2-2}$ -Al <sub>4c-2</sub>	$M_{2-1}-M_{2-2}$	$\Delta E_{binding}$
D(Sc <sub>2</sub> )	2.00	2.01	2.70	2.11	2.06	2.76	2.48	4.65
D(Ti <sub>2</sub> )	1.95	1.96	2.59	2.20	2.34	2.62	2.44	4.19
$D(V_2)$	1.95	2.03	2.62	2.44	3.03	2.83	1.91	3.57
D(Cr <sub>2</sub> )	1.97	1.97	2.52	2.09	2.07	2.58	1.74	2.07
$D(Mn_2)$	1.97	1.99	2.57	2.04	2.95	2.94	2.51	3.02
D(Fe <sub>2</sub> )	1.95	2.02	2.52	2.12	3.07	2.94	2.18	3.19
D(Co <sub>2</sub> )	1.95	2.07	2.42	2.16	3.32	3.26	2.14	3.24
D(Ni <sub>2</sub> )	1.91	2.09	2.48	2.05	3.23	3.22	2.21	3.24
$D(Cu_2)$	1.99	2.01	2.46	2.08	3.22	3.23	2.29	3.33



**Fig. 3.** Variation of  $C-O_a(a)$  and  $C-O_b(b)$  bond distances,  $O_a-C-O_b$  angle (c) and  $M_{2-1}-M_{2-2}$  distance (d) in the  $CO_2$  adsorption configurations formed on  $D(M_2)$  as M varies from Sc to Cu. The pink dotted lines show the C-O bond distance in free  $CO_2$  molecule. The grey dotted lines denote the C-O bond distance and O-C-O angle in free  $CO_3^{2-}$  ion. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of the article.)

The schematic structural model for  $CO_2$  adsorption in  $\eta_2$ configuration is shown in Fig. 2b. In  $\eta_2$ , CO<sub>2</sub> was adsorbed at the interface between M2 and the dry surface. The carbon atom interacts with the metal atoms of the dimer. The O<sub>a</sub> atom binds the surface  $Al_{3c}$  atom whereas the  $O_b$  atom shares one of the metal atoms in  $M_2$ . The adsorbed  $CO_2$  in  $\eta_2$  configuration was also distorted from the free CO<sub>2</sub> molecule. Both the C-O<sub>a</sub> and C-O<sub>b</sub> bonds were elongated, and the C-O<sub>a</sub> bond that interacts with both M<sub>2</sub> and the dry surface was further stretched. The variation of the  $C-O_a$  bond distance in  $\eta_2$  configuration as  $M_2$  varies from  $Sc_2$  to  $Cu_2$ was plotted in Fig. 3a. The variation from the left to the right of the periodic table forms a W-shaped curve. The elongation of the C-O<sub>a</sub> bond for CO<sub>2</sub> adsorption on D(Sc<sub>2</sub>), D(Mn<sub>2</sub>), D(Fe<sub>2</sub>) and D(Cu<sub>2</sub>) is more significant than those on other D(M<sub>2</sub>). The most significant elongation of the C-O<sub>a</sub> bond occurs on D(Cu<sub>2</sub>) where the C-O<sub>a</sub> bond was stretched to 1.57 Å. The C-O<sub>b</sub> bond, which only interacts with M<sub>2</sub> part of the system, was less stretched than that of the C-O<sub>a</sub> bond. In  $\eta_2$  configuration, the adsorbed  $CO_2$  is also in a bent configuration with the  $O_a$ –C– $O_b$  angle of about 108– $123^\circ$  (Fig. 3c). The adsorption of  $CO_2$  in  $\eta_2$  configuration also caused the  $M_{2\text{--}1}\text{--}M_{2\text{--}}$  $_2$  (M = Ti to Fe, Ni and Cu) bond to elongate but the  $Sc_{2-1}$ - $Sc_{2-2}$  bond to shorten (Fig. 3d). We note that the  $Co_{2-1}$ - $Co_{2-2}$  distance was almost the same before and after CO2 adsorption in this configuration.

The schematic structural model for  $\eta_3$  configuration is presented in Fig. 2c. In  $\eta_3$ , CO<sub>2</sub> was adsorbed across an O<sub>2c-3</sub>–Al<sub>4c-3</sub> bridge site away from the metal dimer as a bidentate carbonate species. The O<sub>a</sub> atom was bound to the Al<sub>4c-3</sub> atom whereas the carbon atom was on the O<sub>2c-3</sub> atom. The adsorbed CO<sub>2</sub> in  $\eta_3$  configuration was also distorted from the free CO<sub>2</sub> molecule. In this configuration, the surface is the key factor that induces the distortion of CO<sub>2</sub>. The supported metal dimer has an indirect, secondary effect on CO<sub>2</sub> adsorption and activation through the supporting oxide. As shown in Fig. 3d, the changes in the M<sub>2-1</sub>–M<sub>2-2</sub> bond distance caused by CO<sub>2</sub> adsorption in  $\eta_3$  configuration was insignificant as compared with that in bare D(M<sub>2</sub>).

The adsorption energies for  $CO_2$  adsorption on  $D(M_2)$  are summarized in Table 2. The stability of different adsorption

configurations follows the order of  $\eta_2>\eta_1>\eta_3.$  As such, on  $D(M_2)$ , the most favorable adsorption configuration of  $CO_2$  is  $\eta_2$  in which CO<sub>2</sub> interacts with both M<sub>2</sub> and the support surface directly. The distortion of the adsorbed CO<sub>2</sub> from the free CO<sub>2</sub> molecule in  $\eta_2$  configuration is more significant than those in either  $\eta_1$  or  $\eta_3$ . In Fig. 4, the variation of adsorption energies of CO<sub>2</sub> in different configurations formed on D(M<sub>2</sub>) are plotted against the metals. We also showed the variation of the ionization energies of M<sub>2</sub> as M<sub>2</sub> changes from Sc<sub>2</sub> to Cu<sub>2</sub> in Fig. 4. Ionization energy, which is defined as the energy needed to remove an electron from the neutral M<sub>2</sub>, has been used to measure the electron-donating ability of M<sub>2</sub> [45–52]. Higher ionization energy generally corresponds to lower electron-donating ability. The ionization energy of the dimer increases from Sc<sub>2</sub> to Cu<sub>2</sub>, indicating a decreasing electrondonating ability from left to right across the periodic table. Consequently, CO<sub>2</sub> adsorption energy decreases from the left to the right of the periodic table.

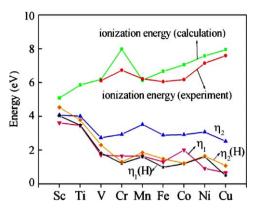
Bader charge analyses were carried out for all adsorption systems. The results are summarized in Table 2. As a reference, the Bader charges for the bare D(M2) without adsorbed CO2 are also included in Table 2. Upon CO<sub>2</sub> adsorption, the electronic charges were redistributed among the constituent atoms: (i) the adsorbed CO<sub>2</sub> became negatively charged; (ii) the surface became less negative whereas the supported M<sub>2</sub> became more positive. These results indicated that there was a charge transfer to the adsorbed CO<sub>2</sub> from both M<sub>2</sub> and the surface. The electron transfer from the substrate, including metal and the oxide support, to adsorbed CO<sub>2</sub> resulted in the activation of CO<sub>2</sub>, as indicated by the stretched C-O distances in adsorbed CO<sub>2</sub> [18,19,21,53-57]. Our results also showed that the electronic charges on adsorbed  $CO_2$  in  $\eta_2$  configuration are greater than those in either  $\eta_1$  or  $\eta_3$  configuration. For example, on D(Ni<sub>2</sub>), the electronic charges on adsorbed  $CO_2$  in  $\eta_1$ ,  $\eta_2$  and  $\eta_3$  are -0.61|e|, -1.13|e| and -0.43|e|, respectively. As such, the adsorbed  $CO_2$  in  $\eta_2$  configuration gained more electronic charge from the substrate and is expected to be more activated than in other configurations. This is consistent with the further stretched C-O<sub>a</sub> bond in  $\eta_2$  configuration, as shown in Fig. 3.

**Table 2** Bader charges (in |e|) for bare  $D(M_2)$  and different  $CO_2$  adsorption configurations on  $D(M_2)$  (M = Sc - Cu) as well as the adsorption energies (in eV) of  $CO_2$  in different adsorption configurations. The Bader charges are the sum of the atomic Bader charges of the respective subset of atoms.

	Bader charges	Adsorption		
	Dry surface	M <sub>2</sub>	Adsorbed CO <sub>2</sub>	energy ( $\Delta E_{ad}$ )
D(Sc <sub>2</sub> )				
Bare	-1.82	1.82		
$\eta_1$	-1.60	2.67	-1.07	3.61
$\eta_2$	-0.66	2.62	-1.96	4.08
$\eta_3$	-1.48	1.97	-0.49	0.54
D(Ti <sub>2</sub> )				
Bare	-1.37	1.37	0.00	0.40
$\eta_1$	-1.22	2.10	-0.88	3.46
$\eta_2$	-0.36	2.06	-1.70	4.01
$\eta_3$	-1.07	1.54	-0.47	0.56
$D(V_2)$				
Bare	-0.93	0.93		
$\eta_1$	-1.01	1.75	-0.74	1.70
$\eta_2$	-0.45	1.56	-1.11	2.73
$\eta_3$	-0.64	1.08	-0.44	0.42
D(Cr <sub>2</sub> )				
Bare	-1.11	1.11		
$\eta_1$	-0.87	1.66	-0.79	1.64
$\eta_2$	-0.25	1.60	-1.35	2.93
$\eta_3$	-0.73	1.19	-0.46	0.35
D(Mn <sub>2</sub> )				
Bare	-0.89	0.89		
$\eta_1$	-0.71	1.47	-0.76	1.64
$\eta_2$	-0.06	1.59	-1.53	3.52
$\eta_3$	-0.50	0.95	-0.45	0.34
D(Fe <sub>2</sub> )				
Bare	-0.64	0.64		
$\eta_1$	-0.54	1.20	-0.66	1.30
$\eta_2$	0.05	1.33	-1.38	2.89
$\eta_3$	-0.26	0.70	-0.44	0.37
D(Co <sub>2</sub> )				
Bare	-0.47	0.47		
$\eta_1$	-0.38	0.97	-0.59	2.00
$\eta_2$	-0.12	0.89	-0.77	2.91
$\eta_3$	-0.08	0.51	-0.43	0.30
D(Ni <sub>2</sub> )				
Bare	-0.37	0.37		
$\eta_1$	-0.20	0.81	-0.61	0.92
$\eta_2$	0.24	0.89	-1.13	3.07
$\eta_3$	0.02	0.41	-0.43	0.32
D(Cu <sub>2</sub> )				
Bare	-0.36	0.36		
$\eta_1$	-0.31	0.78	-0.47	0.66
$\eta_2$	0.21	0.97	-1.18	2.52
$\eta_3$	0.10	0.36	-0.46	0.44

# 3.3. $M_2$ supported on the hydroxylated $\gamma$ - $Al_2O_3$ (1 1 0) surface

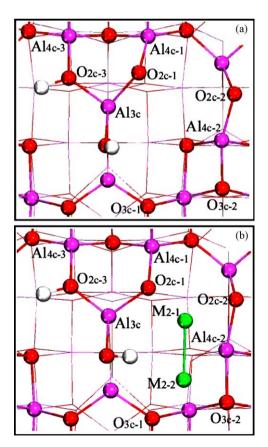
Following the study of  $CO_2$  adsorption over  $D(M_2)$ , we used the hydroxylated  $\gamma$ -Al $_2O_3(1\ 1\ 0)$  surface to characterize the effect of co-adsorbed hydroxyls on  $CO_2$  interaction with the supported metal dimers. The hydroxylated  $\gamma$ -Al $_2O_3$  (1 1 0) surface for supporting  $M_2$  was established in our previous work [33]. As shown in Fig. 5a, on the hydroxylated surface, the proton and hydroxyl group were bound to the surface  $O_{2c-3}$  and  $Al_{3c}$  atoms, respectively, resulting in two OH groups in the unit cell. Again we explored a number of surface sites for  $M_2$  on the hydroxylated surface. The most stable configuration among the configurations for each dimer, denoted as  $H(M_2)$ , was selected as the substrate for  $CO_2$  adsorption. For example,  $H(Ni_2)$  represents the most stable configuration for  $Ni_2$  supported on the hydroxylated  $\gamma$ -Al $_2O_3$  (1 1 0) surface. A schematic structural



**Fig. 4.** Variation of adsorption energy of  $CO_2$  in  $\eta_1$ ,  $\eta_2$ ,  $\eta_1(H)$  and  $\eta_2(H)$  configurations as the function of metal. The calculated and experimentally measured ionization energies of  $M_2$  as the function of metals were also shown. The ionization energies were taken from Refs. [45–52].

model of  $H(M_2)$  is shown in Fig. 5b. The location of the metal dimer in  $H(M_2)$  is similar to that in  $D(M_2)$ . The binding energies of  $M_2$  in  $H(M_2)$  together with the distances between metal and surface atoms are summarized in Table 3.

As shown in Table 3, Sc<sub>2</sub>, with a binding energy of 3.09 eV, shows the strongest binding with the surface among all  $M_2$ . The dimer was adsorbed on the surface through the  $Sc_{2-1}-O_{2c-1}$  (2.08 Å),  $Sc_{2-1}-O_{2c-2}$  (2.10 Å),  $Sc_{2-2}-O_{3c-1}$  (2.20 Å) and  $Sc_{2-2}-O_{3c-2}$  (2.11 Å) bonds. The binding energy of  $Ti_2$  in  $H(Ti_2)$  is 2.94 eV.  $Ti_{2-1}-O_{2c-1}$  (2.11 Å) and  $Ti_{2-1}-O_{2c-2}$  (2.11 Å) bonds were formed between  $Ti_2$  and the hydroxylated surface. The binding energy of  $V_2$  in  $H(V_2)$  is 2.10 eV. In  $H(V_2)$ , the dimer binds the surface only through the



**Fig. 5.** Top view of the hydroxylated  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (1 1 0) surface (a) and schematic structural model of H(M<sub>2</sub>) (b). Color coding: white, H atoms; others are the same as in Fig. 1.

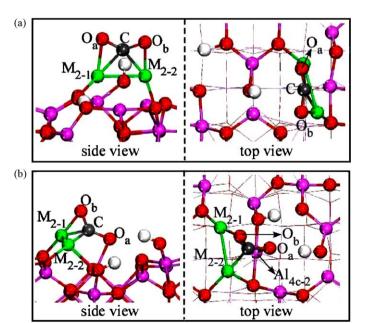
**Table 3** Distances (in  $\dot{A}$ ) between metal and surface atoms and binding energies (in eV) of  $M_2$  in  $H(M_2)$  (M = Sc-Cu).

	M <sub>2-1</sub> -O <sub>2c-1</sub>	M <sub>2-1</sub> -O <sub>2c-2</sub>	M <sub>2-1</sub> -Al <sub>4c-2</sub>	M <sub>2-2</sub> -O <sub>3c-1</sub>	M <sub>2-2</sub> -O <sub>3c-2</sub>	M <sub>2-2</sub> -Al <sub>4c-2</sub>	M <sub>2-1</sub> -M <sub>2-2</sub>	$\Delta E_{binding}$
H(Sc <sub>2</sub> )	2.08	2.10	2.69	2.20	2.11	2.68	2.33	3.09
H(Ti <sub>2</sub> )	2.11	2.11	2.57	2.36	2.33	2.55	2.27	2.94
$H(V_2)$	2.10	2.09	2.68	3.86	3.46	2.79	1.80	2.10
H(Cr <sub>2</sub> )	2.16	2.13	2.68	3.98	3.46	2.82	1.64	1.73
$H(Mn_2)$	2.18	2.30	2.52	2.48	2.18	2.52	2.47	2.03
H(Fe <sub>2</sub> )	2.12	2.63	2.52	2.38	2.24	2.51	2.14	2.36
H(Co <sub>2</sub> )	2.01	2.09	2.47	3.63	2.98	2.56	2.06	2.39
H(Ni <sub>2</sub> )	2.02	2.60	2.48	2.44	2.08	2.37	2.20	2.44
H(Cu <sub>2</sub> )	1.98	2.56	2.59	3.18	2.36	2.40	2.28	2.92

 $V_{2-1}$  atom, with the  $V_{2-1}-O_{2c-1}$  and  $V_{2-1}-O_{2c-2}$  distances of 2.10 Å and 2.09 Å, respectively. The  $V_{2-2}$  atom does not interact with the surface atoms directly. The binding energy of  $Cr_2$  in  $H(Cr_2)$  is lower by 0.37 eV than that of  $V_2$  in  $H(V_2)$ .  $Cr_2$  interacts with the surface through  $Cr_{2-1}-O_{2c-1}$  (2.16 Å) and  $Cr_{2-1}-O_{2c-2}$  (2.13 Å) bonds. From  $Cr_2$  to  $Cu_2$ , the binding energy increases slightly. The binding energy of  $Mn_2$ ,  $Fe_2$ ,  $Co_2$ ,  $Ni_2$  and  $Cu_2$  in  $H(Mn_2)$ ,  $H(Fe_2)$ ,  $H(Co_2)$ ,  $H(Ni_2)$  and  $H(Cu_2)$  are 2.03 eV, 2.36 eV, 2.39 eV, 2.44 eV and 2.92 eV, respectively.  $H(Mn_2)$ ,  $H(Fe_2)$ ,  $H(Ni_2)$  and  $H(Cu_2)$  have a similar structure: the dimer binds the surface through both ends:  $M_{2-1}-O_{2c-1}$  and  $M_{2-2}-O_{3c-2}$ . For instance,  $Mn_{2-1}-O_{2c-1}$  and  $Mn_{2-2}-O_{3c-2}$  in  $H(Mn_2)$  are both 2.18 Å. In  $H(Co_2)$ , the  $Co_{2-1}-O_{2c-1}$  and  $Co_{2-1}-O_{2c-2}$  are 2.01 Å and 2.09 Å, respectively.

# 3.4. $CO_2$ adsorption and activation on $H(M_2)$

In Section 3.2, we showed that  $CO_2$  can be adsorbed across the surface  $O_{2c\text{-}3}$ -Al $_{4c\text{-}3}$  bridge site to form adsorption configuration  $\eta_3$  on  $D(M_2)$ . We also showed that the  $O_{2c\text{-}3}$  and  $Al_{3c}$  sites may be occupied by hydroxyls [33]. Consequently, hydroxylation of the surface is expected to drive  $CO_2$  adsorption to the supported metal dimer, either interacting with the dimer only or at the interface of the dimer and the support. We examined  $CO_2$  adsorption on  $H(M_2)$  and focus our attention on the sites in the vicinity of the metal dimer. Among configurations on  $H(M_2)$  that were explored for  $CO_2$  adsorption, two were found to be stable. The first one has a local structure that is similar to  $\eta_1$  on  $D(M_2)$  and is denoted as  $\eta_1(H)$ ,



**Fig. 6.** Schematic structural models of  $\eta_1(H)$  (a) and  $\eta_2(H)$  (b) configurations. See Figs. 1, 2 and 5 for color coding.

whereas the second one has a similar local structure to that of  $\eta_2$  on  $D(M_2)$  and is referred to as  $\eta_2(H)$ .

A schematic structural model of  $\eta_1(H)$  is shown in Fig. 6a. In  $\eta_1(H)$ , the carbon atom interacts with both metal atoms whereas the  $O_a$  and  $O_b$  atoms bind the  $M_{2-1}$  and  $M_{2-2}$  atoms, respectively. The distortion of adsorbed  $CO_2$  from the free  $CO_2$  molecule is again summarized in a plot, shown in Fig. 7, where the variation of the C-O<sub>a</sub> and C-O<sub>b</sub> distances and the O<sub>a</sub>-C-O<sub>b</sub> angle are plotted as a function of metals. Similar to  $CO_2$  adsorption on  $D(M_2)$ , both  $C-O_a$  and  $C-O_b$  bonds were stretched and the  $O_a-C-O_b$  angle was reduced in  $\eta_1(H)$  configuration (Fig. 7a-c). For instance, in  $\eta_1(H)$ formed upon CO<sub>2</sub> adsorption on H(Ni<sub>2</sub>), the C-O<sub>a</sub> and C-O<sub>b</sub> bonds were stretched to 1.24 Å and 1.30 Å, respectively, and the O<sub>a</sub>-C-O<sub>b</sub> angle was reduced to 130°. As shown in Fig. 7d, adsorption of  $CO_2$  in  $\eta_1(H)$  configuration makes the  $M_{2-1}-M_{2-2}$  distance in supported  $M_2$  with M = Ti to Cu longer whereas the  $Sc_{2-1} - Sc_{2-2}$ distance is shorter than the corresponding  $M_{2-1}-M_{2-2}$  distance prior to CO<sub>2</sub> adsorption.

Fig. 6b shows the schematic structural model of  $\eta_2(H)$  configuration. In  $\eta_2(H)$ ,  $CO_2$  interacts with both the dimer and the surface. The carbon atom binds both metal atoms in  $M_2$  whereas the  $O_a$  and  $O_b$  atoms interact with the surface  $Al_{4c^{-2}}$  atom and one of the metal atoms in  $M_2$ , respectively. The adsorbed  $CO_2$  in  $\eta_2(H)$  configuration was also distorted from the free  $CO_2$  molecule, moreover, the distortion of  $CO_2$  in  $\eta_2(H)$  is more significant than that in  $\eta_1(H)$  (Fig. 7a–c). Compared with the bare  $H(M_2)$ , adsorption of  $CO_2$  in  $\eta_2(H)$  configuration caused the  $M_{2-1}$ – $M_{2-2}$  (M = Sc, Ti, Cr–Cu) distance to elongate and the  $V_{2-1}$ – $V_{2-2}$  bond to shorten, as shown in Fig. 7d.

The adsorption energies for  $CO_2$  on  $H(M_2)$  are summarized in Table 4. The stability of  $\eta_2(H)$  configuration is higher than that of  $\eta_1(H)$  configuration. This again indicates that the interface between  $M_2$  and the surface is more favorable for  $CO_2$  adsorption on  $H(M_2)$ . We plotted the adsorption energies of  $CO_2$  in  $\eta_1(H)$  and  $\eta_2(H)$  configurations as a function of metals in Fig. 4. As shown in Fig. 4, the strength of the interaction between  $CO_2$  and  $H(M_2)$  becomes weaker as  $M_2$  changes from  $Sc_2$  to  $Cu_2$ .

Bader charges were calculated for all adsorption configurations formed upon  $CO_2$  adsorption on  $H(M_2)$  and are summarized in Table 4. The Bader charges for the bare  $H(M_2)$  are also given as references. Again, a charge transfer from both the dimer and the support to the adsorbed  $CO_2$  occurred. The charge accumulation in adsorbed  $CO_2$  resulted in activation of  $CO_2$ . Furthermore, the charges on the adsorbed  $CO_2$  in  $\eta_2(H)$  configuration are greater than that in  $\eta_1(H)$  configuration for the same metal, indicating that the adsorbed  $CO_2$  gained more electrons from the substrate in  $\eta_2(H)$  configuration. The greater electronic transfer led to a more pronounced distortion of  $CO_2$  in  $\eta_2(H)$  configuration.

# 3.5. General discussion

As we noted in previous sections,  $M_2$  was adsorbed at a position that is surrounded by  $O_{2c}$ ,  $O_{3c}$ ,  $Al_{3c}$  and  $Al_{4c}$  atoms on both the dry

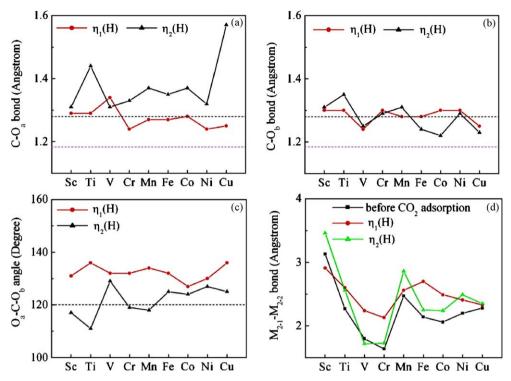


Fig. 7. Variation of  $C-O_a(a)$  and  $C-O_b(b)$  bond distances,  $O_a-C-O_b$  angle (c) and  $M_{2-1}-M_{2-2}$  distance (d) in different adsorption configurations formed on  $H(M_2)$  as a function of metals. The pink and grey dotted lines are for free  $CO_2$  and  $CO_3^{2-}$ , as shown in Fig. 3. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of the article.)

and hydroxylated  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (1 1 0) surfaces. Furthermore, the binding energy of the dimer on the dry surface varies in a similar trend to that on the hydroxylated surface: the binding energy first decreases from Sc<sub>2</sub> to Cr<sub>2</sub>, and then increases slowly from Cr<sub>2</sub> to Cu<sub>2</sub>. As shown in Tables 1 and 3, the binding energy of M<sub>2</sub> in D(M<sub>2</sub>) is greater than that in H(M<sub>2</sub>), indicating that hydroxylation of the surface weakens the binding of M<sub>2</sub> to the surface.

As shown in Fig. 1a, there are two  $O_{2c}$ -Al<sub>3c</sub>  $(O_{2c-1}$ -Al<sub>3c</sub> and  $O_{2c-3}-Al_{3c}$ ) and three  $O_{2c}-Al_{4c}$  ( $O_{2c-1}-Al_{4c-1}$ ,  $O_{2c-2}-Al_{4c-2}$  and  $O_{2c-1}$  $_3$ -Al $_{4c-3}$ ) sites on the clean dry  $\gamma$ -Al $_2$ O $_3$  (1 1 0) surface. In our previous work [33], we demonstrated that the  $O_{2c}$ -Al<sub>3c</sub> ( $O_{2c-1}$ -Al<sub>3c</sub> or O<sub>2c-3</sub>-Al<sub>3c</sub>) bridge site is the most favorable site for CO<sub>2</sub> adsorption on the clean dry surface. CO2 can also be adsorbed at the  $O_{2c}$ - $Al_{4c}$  ( $O_{2c-1}$ - $Al_{4c-1}$ ,  $O_{2c-2}$ - $Al_{4c-2}$  or  $O_{2c-3}$ - $Al_{4c-3}$ ) bridge site, but the binding strength of CO2 at the O2c-Al4c site is weaker than that at the O<sub>2c</sub>-Al<sub>3c</sub> site. The adsorption energies for CO2 adsorption at the O2c-Al3c and O2c-Al4c bridge sites of the clean dry surface were 0.43 eV and 0.27 eV, respectively. On the hydroxylated  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (1 1 0) surface, the O<sub>2c-3</sub> and Al<sub>3c</sub> sites were occupied by hydroxyls (Fig. 5a). Due to the occupation of the  $O_{2c-3}$  and  $Al_{3c}$  sites by hydroxyls, the  $O_{2c}-Al_{4c}$  ( $O_{2c-1}-Al_{4c-1}$ or O<sub>2c-2</sub>-Al<sub>4c-2</sub>) bridge site became favorable for CO<sub>2</sub> adsorption on the hydroxylated surface. The adsorption energy for CO2 adsorption at the O<sub>2c</sub>-Al<sub>4c</sub> bridge site of the hydroxylated surface was 0.20 eV [33].

Among three  $CO_2$  adsorption configurations,  $\eta_1$ ,  $\eta_2$  and  $\eta_3$ , the adsorption energies of  $CO_2$  in  $\eta_1$  and  $\eta_2$  are clearly much higher than that of  $CO_2$  on the dry surface [33]. Even in  $\eta_3$  configuration where  $CO_2$  was adsorbed at a surface  $O_{2c}$ – $Al_{4c}$  bridge site, the adsorption energy on  $D(M_2)$  is greater than that of  $CO_2$  adsorption at the  $O_{2c}$ – $Al_{4c}$  site of the clean dry surface. Two adsorption configurations examined on  $H(M_2)$ ,  $\eta_1(H)$  and  $\eta_2(H)$ , are similar to  $\eta_1$  and  $\eta_2$ , respectively. The adsorption energies for  $CO_2$  in  $\eta_1(H)$  and  $\eta_2(H)$  are also much greater than that of  $CO_2$  on the hydroxylated surface. As such, the interaction between  $CO_2$  and  $H(M_2)$  is much stronger than that of  $CO_2$  with the hydroxylated

surface. These results illustrated that the presence of M<sub>2</sub> enhances the binding strength of the substrate toward CO<sub>2</sub>.

Following Ref. [24], the adsorption of  $CO_2$  on  $M_2/\gamma$ -Al<sub>2</sub>O<sub>3</sub> can be considered as a two-step process: first, CO<sub>2</sub> deforms from its linear gas phase structure into a bent CO<sub>2</sub> fragment; second, the bent CO<sub>2</sub> fragment binds  $M_2/\gamma$ -Al<sub>2</sub>O<sub>3</sub>. The process is shown schematically in Fig. 8. We defined the energy cost for the first step as distortion energy and the energy gain for the second step as binding energy. The sum of the energies of these two steps yields the adsorption energy. In Fig. 9a, we plot the distortion energy of  $CO_2$  in  $\eta_2$  and  $\eta_2(H)$  against the electronic charges on adsorbed  $CO_2$ . As shown in the plot, there is approximately a linear relationship between the distortion energy and the electronic charges in both  $\eta_2$  and  $\eta_2(H)$ configurations. Furthermore, the two lines describing the linear relationship in  $\eta_2$  and  $\eta_2(H)$  configurations are almost coincident. The variation of the binding energies as a function of the electronic charges on adsorbed  $CO_2$  in  $\eta_2$  and  $\eta_2(H)$  configurations is plotted in Fig. 9b. Again, linear correlation between the binding energy and the charges on the adsorbed  $CO_2$  in both  $\eta_2$  and  $\eta_2(H)$ configurations exists. However, we note that the slope of binding energy for  $\eta_2(H)$  configuration is greater than that for  $\eta_2$ configuration. This can be attributed to a greater change in electron-donating ability for the early transition metal than for the late ones due to hydroxylation. Generally, hydroxylation reduced the amount of charge transferred to CO<sub>2</sub>, as shown in Fig. 9. Consequently, hydroxylation compresses the range of charge transfer across the periodic table. We note, however, hydroxylation increases the electronic charges on adsorbed  $CO_2$  in  $\eta_2(H)$ configuration for  $Co_2/\gamma$ -Al<sub>2</sub>O<sub>3</sub>.

Improving the adsorption and activation of  $CO_2$  on the catalysts is expected to benefit the conversion of  $CO_2$ , and therefore, the activity and selectivity. Our present results showed that the interface between the metal dimer and the  $\gamma$ -Al $_2O_3$  support is the most favorable site for  $CO_2$  adsorption and  $CO_2$  becomes activated upon adsorption at this site. This can be achieved in practice by increasing the dispersion of metal on the support. For the same

**Table 4** Bader charges (in |e|) for bare  $H(M_2)$  and different  $CO_2$  adsorption configurations on  $H(M_2)$  (M = Sc - Cu) as well as the adsorption energies (in eV) of  $CO_2$  in different adsorption configurations. The Bader charges are the sum of the atomic Bader charges of the respective subset of atoms.

	Bader charges	Adsorption		
	Hydroxylated surface	$M_2$	Adsorbed CO <sub>2</sub>	energy ( $\Delta E_{ad}$ )
H(Sc <sub>2</sub> )				
Bare	-1.54	1.54		
$\eta_1(H)$	-1.29	2.45	-1.16	4.03
$\eta_2(H)$	-0.97	2.32	-1.35	4.53
H(Ti <sub>2</sub> )				
Bare	-1.02	1.02		
$\eta_1(H)$	-0.94	1.90	-0.96	3.48
$\eta_2(H)$	-0.38	1.93	-1.55	3.79
$H(V_2)$				
Bare	-0.55	0.55		
$\eta_1(H)$	-0.64	1.50	-0.86	1.83
$\eta_2(H)$	-0.16	1.20	-1.04	2.30
H(Cr <sub>2</sub> )				
Bare	-0.50	0.50		
$\eta_1(H)$	-0.38	1.25	-0.87	1.21
$\eta_2(H)$	-0.05	1.22	-1.17	1.23
H(Mn <sub>2</sub> )				
Bare	-0.75	0.75		
$\eta_1(H)$	-0.49	1.37	-0.88	1.60
$\eta_2(H)$	-0.19	1.52	-1.33	1.86
H(Fe <sub>2</sub> )				
Bare	-0.54	0.54		
$\eta_1(H)$	-0.36	1.19	-0.83	0.99
$\eta_2(H)$	0.08	1.11	-1.03	1.48
• • •				
H(Co <sub>2</sub> ) Bare	-0.35	0.35		
$\eta_1(H)$	-0.35 -0.15	0.55	-0.81	1.20
$\eta_1(H)$	-0.04	0.94	-0.98	1.22
• • •	0.01	0.0 1	0.00	1.22
H(Ni <sub>2</sub> )	0.04	0.04		
Bare	-0.31	0.31	0.61	1.00
$\eta_1(H)$ $\eta_2(H)$	-0.09 0.11	0.70 0.86	-0.61 -0.97	1.60 1.64
•	0,11	0.00	-0.37	1.04
H(Cu <sub>2</sub> )				
Bare	-0.25	0.25	0.55	0.50
$\eta_1(H)$	-0.14	0.71	-0.57	0.50
$\eta_2(H)$	0.06	0.87	-0.93	1.09

metal loading, a large particle will lead to a reduced interface, therefore, low activity. Increasing the dispersion of the metal particles and maintaining the smaller size of supported metal particles will maximize the interface available for CO<sub>2</sub> adsorption. Furthermore, CO<sub>2</sub> adsorbed at the interface is more activated, manifested by the stretched C–O bond. The activated CO<sub>2</sub> will make the subsequent reaction facile. These analyses are consistent with the experimental observation that the catalysts with smaller

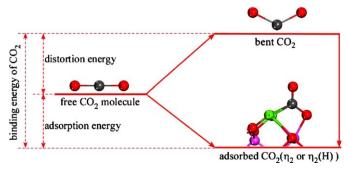


Fig. 8. Schematic of distortion and binding steps of CO<sub>2</sub>.

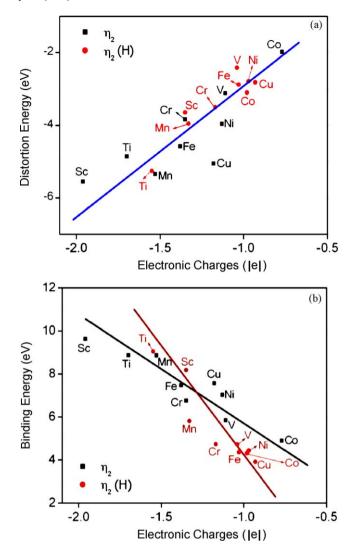


Fig. 9. Variation of the distortion energy (a) and binding energy (b) of  $CO_2$  in  $\eta_2$  and  $\eta_2(H)$  configurations as a function of the electronic charges on adsorbed  $CO_2$ .

metal particles and higher particle dispersion show a high activity in the catalytic conversion of  $CO_2$ .

# 4. Conclusions

The adsorption and activation of  $CO_2$  on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>-supported 3d transition metal dimers  $(M_2/\gamma-Al_2O_3, M=Sc-Cu)$  have been studied using DFT slab calculations. CO2 was found to adsorb on M<sub>2</sub>/γ-Al<sub>2</sub>O<sub>3</sub> negatively charged and in bent configuration, indicating a partial activation of CO<sub>2</sub>. Both the supported metal dimer and the support, either dry or partially hydroxylated γ-Al<sub>2</sub>O<sub>3</sub>, contribute to the activation of CO<sub>2</sub> by donating their electronic charges to the adsorbed CO<sub>2</sub>. The most favorable adsorption site for  $CO_2$  on  $M_2/\gamma$ - $Al_2O_3$ , either dry or partially hydroxylated support, is the interface between the metal dimer and the support. The electronic charges on the adsorbed CO<sub>2</sub> make CO<sub>2</sub> distort from its linear configuration in the gas phase. The presence of the metal dimer enhances the binding of CO<sub>2</sub> with the substrate, including both dry and partially hydroxylated  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>. As such, the adsorption energy of  $CO_2$  on  $M_2/\gamma$ - $Al_2O_3$  is much higher than that on the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> surfaces without supported metal dimer. We also demonstrated that the binding strength of  $CO_2$  with  $M_2/\gamma$ -Al<sub>2</sub>O<sub>3</sub> correlates with the electron-donating ability of the substrate. Both distortion energy and CO<sub>2</sub> binding energy show

a linear dependence on the amount of charge on the adsorbed CO<sub>2</sub>. The 3d metal on the left of the periodic table has a greater electrondonating ability which resulted in a stronger CO2 binding. Hydroxylation of the support surface reduces the amount of charge transferred to CO<sub>2</sub> for the same metal dimer and weakens the CO<sub>2</sub> chemisorption bonds. Highly dispersed metal particles maintained at a small size are expected to exhibit good activity toward CO<sub>2</sub> adsorption and activation.

# Acknowledgements

We gratefully acknowledge support from the Petroleum Research Fund (PRF-G44103-G10), the National Natural Science Foundation of China (under contract 20490203), and the Illinois Clean Coal Institute.

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